# Decoration of Electrospun Nanofibers with Magnetic Nanoparticles via Electrospinning and Sol-gel Process

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**Abstract** Polyacrylonitrile(PAN)/Fe<sub>3</sub>O<sub>4</sub> composite nanofibers were successfully obtained through electrospinning and sol-gel technology. The resulting magnetic  $Fe_3O_4$  nanoparticles were homogeneously distributed on the surface of PAN nanofibers and the diameters of polyacrylonitrile nanofibers and nanoparticles were easily controlled, respectively. The distribution of  $Fe_3O_4$  nanoparticles inside the nanofibrous composite was investigated by field emission scanning electron microscopy and transmission electron microscopy. X-ray diffraction reveals the presence of  $Fe_3O_4$ nanoparticles in the composite nanofiber. The resulting sample shows a super paramagnetic behavior.

Keywords Electrospinning; Magnetic; Fe<sub>3</sub>O<sub>4</sub>; Nanoparticle; Sol-gel

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## 1 Introduction

Recently, nanoscale magnetic particles have attracted a great interest in fundamental and application researches with special emphasis on their size dependent properties<sup>[1]</sup>. Among magnetic nanoparticles, Fe<sub>3</sub>O<sub>4</sub> has become the focus due to its magnetic, catalytic, conductive and biological properties which could be applied in many fields including magnetic storage devices<sup>[2]</sup>, ferrofluids<sup>[3]</sup>, magnetic carriers for drug targeting<sup>[4]</sup>, the separation of biochemical products<sup>[5]</sup> and catalysis<sup>[6]</sup>. For these purposes, well-dispersed chemically stable nanoparticles with uniform size and shape are desired. In this respect, the synthesis of composite nanomaterials where the magnetic nanoparticles are integrated into organic or inorganic matrix is particularly promising since it allows not only the control over the particle growth but also the spatial arrangement in some cases. Polymers<sup>[7]</sup>, glass<sup>[8]</sup> or ceramic materials<sup>[9]</sup> have been extensively used as matrix in the preparation of nanocomposites containing Fe<sub>3</sub>O<sub>4</sub> nanoparticles. These composites would be a perfect candidate for heterogeneous catalysis with excellent thermal and chemical stability and good magnetic performance.

Electrospinning technique<sup>[10-14]</sup> has been established as a unique and cost effective approach for fabricating fibers with large surface area for various application. Electrospun nanofibers are good matrices to support nanoparticles. A lot of work has been done about the synthesis of nanoparticles/electrospun nanofiber composites. Li et al.<sup>[15]</sup> provided a simple method for fabricating size controllable single-crystal Ag nanoparticles on the surface of polyacrylonitrile(PAN) nanofibers via electrospinning followed by UV irradiation. Dong et al.<sup>[16]</sup> have illustrated the synthesis of gold and silver nanoparticles on the surface of electrospun Poly(4-vinylpyridine)(P4VP) fibers P4VP/poly(methyl methacrylate)(PMMA) composite fibers by absorption of metal ions from a colloidal gold solution and subsequent reduction. Li et al.<sup>[17]</sup> demonstrated that nanostructures can be selectively deposited on electrospun anatase nanofibers via the photocatalytic feature of titania. Di et al.<sup>[18]</sup> manufactured magnetic PAN/Fe<sub>3</sub>O<sub>4</sub> nanocomposite fibers by electrospinning PAN/DMF solution with suspended Fe<sub>3</sub>O<sub>4</sub> nanopaticles, they also examined the influnece of the operating parameters and the rheologial behavior of the solution. However, they did not show how the particles were dispersed inside the nanofibers and using this method is hard to fabricate well dispersed magnetic nanoparticles/nanofibers composites with their surface homogeneously covered by

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mono-dispersed metal nanoparticles with controllable size because the major difficulty to manufacture magnetic nanoparticles is their undesirable agglomeration to form larger particles.

In this study, we demonstrated the synthesis of ultrafine PAN fibers containing mono-dispersed  $Fe_3O_4$  nanoparticles *via* electrospinning followed by a sol-gel process, the morphological characteristics and magnetic properties of the resulting nanofibrous composites were characteraized by scanning electron microscopy, X-ray diffraction, transmission electron microscopy and vibrating sample magnetometry.

#### 2 Experimental

PAN with a relative molecular mass of 150000 was purchased from Aldrich Co. N,N-dimethylformarmide(DMF), ferrous chloride(FeCl<sub>2</sub>·4H<sub>2</sub>O), ferric chloride(FeCl<sub>3</sub>·6H<sub>2</sub>O), hydrochloric acid(HCl), sodium hydroxide(NaOH) were purchased from Tianjin Tiantai Fine Chemicals Co., all the reagents were analytical grade and used as received without further purification.

Fibers of PAN/FeCl3 were obtained by electrospin via an apparatus and process described previously<sup>[15]</sup>. PAN/DMF solution was prepared by dissolving 10% PAN and 2%(mass fraction) FeCl<sub>3</sub>(the molar ratio of  $Fe^{3+}$ : AN =1:30) in DMF for 6 h under 90 °C. During electrospinning, the applied voltage was held constant at 15 kV, and the distance between the syringe and the aluminum foil collector was kept at 25 cm. After 0.2 mL of 12.1 mol/L HCl and 25 mL of purified, deoxygenated water(by nitrogen gas bubbling for 30 min) were mixed, 1.44 g(0.0053 mol) of FeCl<sub>3</sub>·6H<sub>2</sub>O and 0.55 g(0.00267 mol) of FeCl<sub>2</sub>·4H<sub>2</sub>O were dissolved in the solution with stirring. A piece of 0.075 g electrospun PAN/FeCl<sub>3</sub> mat was cut into several tiny pieces and immerged in the solution for 3 h. The resulting solution with fiber mats suspended in was added dropwise to 250 mL of 1.5 mol/L NaOH/ deoxygenated water solution under vigorous stirring, an instant black precipitate was generated in the solution during this step. The paramagneticity was checked in situ by placing a magnet near the black precipitate of Fe<sub>3</sub>O<sub>4</sub>. When the reaction was completed, the mats were rinsed by plenty of distilled water, followed by a 30 min ultrasonic process.

The PAN/FeCl<sub>3</sub> nanofibers were analyzed by scanning electron microscopy(SEM)(SSX-550, Shi-

madzu). A drop of Fe<sub>3</sub>O<sub>4</sub> nanoparticle sol was carefully placed on the copper grid surface and dried for transmission electron microscopy(TEM, Hitachi-600) study. X-ray diffraction(XRD) patterns of the product were recorded on a Rigaku D/max 2500/PC diffractometer with Cu  $K\alpha$  radiation at a tube voltage and current of 40 kV and 200 mA, respectively. The room-temperature magnetization in an applied magnetic field was performed on a model JDM-13 vibrating sample magnetometer.

### 3 Results and Discussion

Fig.1 shows SEM and TEM images of PAN/ FeCl<sub>3</sub> electrospun nanofibers under magnifications. Fig.1(A) is a low-magnification view of the nanofibers[the molar ratio of Fe<sup>3+</sup>:AN<sup>-</sup>(Cyano) is 1:30], showing the electrospun nanofibers have smooth morphology with an average diameter of about 150 nm, and the nanofibers diameter could be well controlled by adjusting electrospinning parameters such as polymer concentration and the operational condi tions such as the strength of applied electric field. During the process of Fe<sub>3</sub>O<sub>4</sub> formation in solution, the color of PAN/FeCl<sub>3</sub> nanocomposite fibers gradually changed from yellow to black, indicating that  $Fe^{3+}$  on the surfaces of PAN nanofibers reacted with Fe<sup>2+</sup> in the solution and to form small aggregated Fe<sub>3</sub>O<sub>4</sub> nanoparticles. Fig.1(B) is a typical TEM image of as-prepared product, showing that the surfaces of the PAN nanofibers are homogeneously covered by uniform Fe<sub>3</sub>O<sub>4</sub> nanoparticles with an average diameter of 8 nm, the inset shows the electron diffraction pattern(ED) image of Fe<sub>3</sub>O<sub>4</sub> nanoparticles, which confirms that the oxide is crystal and has an inverse cubic spinal structure. Because of the salts distributed simultaneously within the polymer and in the solution, the nanoparticles are easily dispersed homogeneously in the circumstance and the sizes could be preciously controlled by the sol-gel process which had been intensely investigated<sup>[19]</sup>.

The size distribution of the formed particles in the solution and on the surface of polymer fibers are quite the same, indicating a homogeneous atmosphere between polymer-inside and polymer-outside[see Fig.1(D)], so we assumed that each  $Fe^{3+}$  on the surface of polymer nanofiber reacted with another  $Fe^{3+}$  and a  $Fe^{2+}$  in the solution according to reaction(1), just as the other  $Fe^{3+}$  in the solution, forming  $Fe_3O_4$   $FeCl_2+2FeCl_3+8NaOH \rightarrow Fe_3O_4+8NaCl+4H_2O$  (1)

To verify this mechanism, a piece of pure electrospun PAN mat was used instead of PAN/FeCl<sub>3</sub> electrospun nanofibers to repeat the reaction as in the experiment. The TEM result[Fig.1(C)] shows that the formed  $Fe_3O_4$  nanoparticles did not grow on the polymer surface, the formed  $Fe_3O_4$  nanoparticles have not any reaction with PAN fibers, indicating the PAN surface-polymer-chains(or the cyanos) extended to the solution phase can not adsorb the  $Fe_3O_4$  nanoparticles onto the polymer surface, which demonstrates that  $Fe_3O_4$  nanoparticles immobilized on the fibers were formed from the  $Fe^{3+}$  which were doped in the PAN fibers.



Fig.1 SEM images of PAN/FeCl<sub>3</sub> nanofibers(A) and TEM images of PAN nanofibers and Fe<sub>3</sub>O<sub>4</sub> napoparticles[(B)—(F)]

(A) The molar ratio of  $Fe^{3+}$ :AN is 1:30; (B) PAN nanofibers with uniform, mono-dispersed  $Fe_3O_4$  nanoparticles[electrospun nanofibers with the molar ratio of  $Fe^{3+}$ :AN=1:30]; (C) PAN nanofibers without  $Fe_3O_4$  nanoparticles growing on their surfaces; (D)  $Fe_3O_4$  nanoprticles forming in solution; (E) PAN nnofibers with  $Fe_3O_4$  nanoprticles on their surfaces. Electrospun nanofibers with the molar ratio of  $Fe^{3+}$ :AN=1:5; (F) electrospun nonfibers with the molar ratio of  $Fe^{3+}$ :AN=1:100.

In our experiment, the sizes and distribution density of the Fe<sub>3</sub>O<sub>4</sub> nanoparticles can be adjusted by varying the molar ratio of Fe<sup>3+</sup> to AN in the electrospinning process. If the molar ratio of Fe<sup>3+</sup>:AN was 1:5, the sizes of the Fe<sub>3</sub>O<sub>4</sub> nanoparticles would be enlarged to 15 nm and aggregated as shown in Fig.1(E), and the distribution density would be much thicker than that of the sample with a molar ratio of 1:30 of Fe<sup>3+</sup>:AN. When the molar ratio of Fe<sup>3+</sup>:AN<sup>-</sup> was lowered to 1:100, scattered mono-dispersed Fe<sub>3</sub>O<sub>4</sub> nanoparticles with an average diameter of 4 nm could be obtained on the surface of PAN nanofibers[Fig.1(F)].

The XRD patterns of Fe<sub>3</sub>O<sub>4</sub> nanoparticles and Fe<sub>3</sub>O<sub>4</sub>/PAN composites are shown in Fig.2. The diffraction peaks of the Fe<sub>3</sub>O<sub>4</sub> nanoparticles(Fig.2 pattern *a*) were measured to be  $2\theta$ =30.38°, 35.62°, 43.18°,

53.45°, 57.08, and 62.84°, which can be assigned to (220), (311), (400), (422), (511) and (440), respectively. These data are in good agreement with those of Fe<sub>3</sub>O<sub>4</sub>(JCPDS No.). The XRD patterns of Fe<sub>3</sub>O<sub>4</sub>/PAN composites(Fig.2*b*) show a broad non-crystalline peak  $(2\theta=20^{\circ}-30^{\circ})$  and a crystalline peak $(2\theta=18^{\circ})$ 



Fig.2 XRD patterns of pure Fe<sub>3</sub>O<sub>4</sub> nanoparticles(*a*), and Fe<sub>3</sub>O<sub>4</sub>/PAN composites(*b*)

corresponding to the orthorhombic PAN (110) reflection<sup>[20]</sup>, and besides the diffraction peaks of the PAN phase, another six peaks appeared corresponding to the above peaks, indicating that the Fe<sub>3</sub>O<sub>4</sub> nanoparticles on the PAN nanofibers have the same crystal diffraction as pure Fe<sub>3</sub>O<sub>4</sub>. On the other hand, the broadening of these peaks suggests the presence of Fe<sub>3</sub>O<sub>4</sub> nanoparticles.

Fig.3 shows the dependence of the magnetization on the applied magnetic fields for  $Fe_3O_4$  nanoparticles and  $Fe_3O_4$ /PAN composite at room temperature. The observation of hysteresis loop indicating this superparamagnetic behavior is similar to that of  $Fe_3O_4$  nanoparticles. Because the magnetic properties are contributing to the  $Fe_3O_4$  nanoparticles in the composites, it is reasonable to believe that the superparamagnetic behavior of the composites can be enhanced by increasing the doping concentration of  $Fe_3O_4$  nanoparticles.



Fig.3 Magnetization *versus* applied magnetic field at room temperature for Fe<sub>3</sub>O<sub>4</sub> nanoparticles(*a*), Fe<sub>3</sub>O<sub>4</sub>/PAN composites(*b*)

#### 4 Conclusions

We have demonstrated the synthesis of ultrafine PAN fibers containing monodisperse  $Fe_3O_4$  nanoparticles *via* electrospinning followed by a sol-gel process. Magnetic  $Fe_3O_4$  nanoparticles on the PAN nanofibers are almost uniform and most of  $Fe_3O_4$  nanoparticles are approximately spherical with a diameter of about 8 nm. The magnetic  $Fe_3O_4$  nanoparticles

with controllable sizes are homogeneously distributed on the surface of the PAN nanofibers. The diameters of the PAN nanofibers and  $Fe_3O_4$  nanoparticles could be controlled by varying the electrospinning and sol-gel process parameters, respectively. The nanocomposite fibers have a large ratio of surface area to volume and the PAN nanofibers have superior mechanical properties and also have super paramagnetic behavior which would find a lot of application.

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