

Short communication

Magnesium-assisted formation of metal carbides and nitrides from metal oxides

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ABSTRACT

A series of carbides (TiC, V₂C, Mo₂C) were synthesized by the corresponding metal oxides (TiO₂, V₂O₅, MoO₃), CaC₂ and magnesium as starting materials in a stainless steel autoclave at 600 °C. Through similar processes, transition metal nitrides (TiN, VN and CrN) could also be produced by employing the corresponding metal oxides (TiO₂, V₂O₅, Cr₂O₃), NaNH₂, and magnesium as starting materials at 550 °C. The FE-SEM image showed that the TiC sample was mainly consisted of flower-like microstructures. TEM image showed the other carbides (V₂C, Mo₂C) and the obtained nitride (TiN, VN, CrN) were consisted of nanoparticles. The possible synthesis mechanism of TiC had been described.

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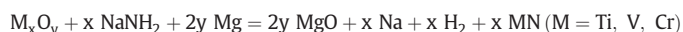
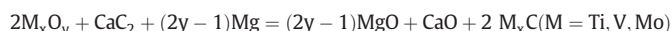
1. Introduction

Metal carbides and nitrides have a wide range of applications in many fields due to their superior properties such as high melting point, extreme hardness, and high resistance to oxidation and corrosion [1,2]. For example, titanium carbide can be used as cutting tools and reinforcing components in composites [3]. Vanadium carbide is proved to be an active ammonia decomposition catalyst [4]. Molybdenum carbide is a powerful catalyst and considered as the potential substitute of noble metal in catalysis [5]. Titanium nitride can be used not only as abrasion resistant coatings but also as a catalyst for the alkylation of ketones with alcohols [6]. Vanadium nitride is a hard refractory material and used as electrodes in supercapacitors [7]. Chromium nitride can be used to coat steels for corrosion protection [8].

Traditionally, metal carbides and nitrides have been successfully synthesized by direct carbonation/nitridation of the elements (650–1900 °C) [9,10]; carbonation/nitridation of the metal oxides at high-temperature (>800 °C) [11,12]; thermal decomposition of precursors at relatively low temperature [13–15]; solid state metathesis (SSM) [16–18]; chemical vapor deposition (650–1200 °C) [19,20]; solvothermal synthesis [21–23]. The ball milling can be also used to synthesize transition metal carbides [24]. Nitrides can be prepared by ammonolysis of binary metal chlorides (>550 °C) [25], or metal sulfides (800 °C) [26]. Metal carbides and nitrides can be obtained by a soft urea pathway using urea as carbon/nitrogen source at 800 °C [27].

In this study, a magnesium-assisted approach was applied to synthesize transition metal carbides and nitrides. A series of carbides and

nitrides were prepared from the corresponding metal oxides *via* a magnesium-assisted solid-state reaction. The reactions *via* the route can be indicated as:



2. Experimental procedure

2.1. Preparation of metal carbides

All chemical reagents used in this experiment were of analytical grade and used without further purification. In a typical synthesis process, transition metal oxides (2 mmol of TiO₂, 1 mmol of V₂O₅ or 2 mmol of MoO₃), 1 mmol of calcium carbide and 50 mmol of metallic magnesium were mixed in an agate mortar. The mixture was sealed in a stainless steel autoclave (20 mL). The temperature of the electronic furnace was raised from room temperature to 600 °C with a heating ramp rate of 10 °C/min and kept at 600 °C for 10 h, and then it was cooled to room temperature naturally. A dark precipitate was collected and washed with dilute hydrochloric acid under ultrasonic, absolute ethanol, and distilled water. The final product was dried in vacuum at 60 °C for 5 h.

2.2. Preparation of metal nitrides

In a typical synthesis process, metal oxides (2 mmol of TiO₂, 1 mmol of V₂O₅, or 1 mmol of Cr₂O₃), 0.1 mol of sodium amide and an excess of 20 mmol of metallic magnesium were mixed and placed in a stainless-steel autoclave of 20 mL capacity. The autoclave was sealed and put into an electronic furnace which was heated from room temperature at 10 °C/min to 550 °C and then maintained for 10 h. After that, the

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autoclave was cooled to room temperature naturally. In order to remove the byproducts, the precipitates in the autoclave were collected and washed with dilute hydrochloric acid, absolute ethanol, and distilled water several times. After that, the products were dried in vacuum at 60 °C for 5 h for further characterization.

2.3. Characterization

Powder X-ray diffraction (XRD) measurements were carried out with a Philips X'pert X-ray diffractometer ($\text{CuK}\alpha$ $\lambda = 1.54178$ Å). X-Ray photoelectron spectra (XPS) were recorded on a VGESCALAB MKII X-ray photoelectron spectrometer, using non-monochromated Mg K α X-ray radiation as the excitation source. The scanning electron microscopy (SEM) images were taken by using a field-emitting scanning electron microscope (FESEM, JEOL-JSM-6700F). The transmission electron microscopy (TEM) images, high-resolution transmission electron microscopy (HRTEM) images and the selected-area electron diffraction (SAED) patterns were taken on a JEOL-2010 transmission electron microscope with an accelerating voltage of 200 kV.

3. Results and discussion

3.1. Characterization of the synthesized TiC

X-Ray powder diffraction (XRD) analysis was used to determine the phases of the samples. Fig. 1a shows the typical XRD pattern of the TiC sample. All of the peaks could be indexed to cubic TiC with lattice constant of $a = 4.3256$ Å, which is close to the value of cubic TiC ($a = 4.3276$ Å) (Joint Committee on Powder Diffraction Standards (JCPDS) card no. 73-0472). No other crystalline impurities were detected by XRD, which indicated the pure cubic TiC sample was obtained via the present synthetic route. FE-SEM, TEM, and HRTEM images of the TiC sample are shown in Fig. 1b–d. The FE-SEM image in Fig. 1b shows that the sample is mainly consisted of flower-like microstructures together with a small fraction of the irregular particles.

The diameter of microstructures ranges from 0.5 to 1 μm . The FE-SEM image indicates that the flower-like microstructures are composed of nanosheets. The microstructure of the sample was further determined with TEM, which prove the result of the FE-SEM (was shown in Fig. 1c). Moreover, the selected area electron diffraction pattern (inset of Fig. 1c) can be indexed to the 111 and 200 reflections in cubic TiC consistent with the crystal viewed down the $[01\bar{1}]$ axis. The HRTEM image of a nanosheet shown in Fig. 1d presents the lattice spacing. The distance separating the lattice fringes, around 0.25 nm, coincides with the distance between two (111) planes in cubic TiC.

The XPS spectrum of as-prepared TiC was also investigated (Fig. 2). As can be seen from the curve a and b, three strong peaks centered at about 281.50, 454.71, and 460.68 eV correspond to the C1s, Ti2p3/2, and Ti2p1/2 binding energy of TiC in literature [28]. An average composition of Ti: C is 1.12: 1 which could be quantified by the Ti2p and C1s peak areas.

3.2. Characterization of the synthesized V_2C and Mo_2C

The XRD patterns of V_2C (upper in Fig. 3a) and Mo_2C (bottom in Fig. 3a) samples were shown in the Fig. 3a. The sharp peaks with strong diffraction intensity in the Fig. 3a can be appropriately indexed to hexagonal V_2C (JCPDS card no. 73-1320) and hexagonal Mo_2C (JCPDS card no. 35-0787). The TEM image of V_2C sample (Fig. 3b) reveals the sample is composed of nanoparticles with the average size of 50 nm. The SAED rings (upper-right in Fig. 3b) correspond to (100), (002), (101), (102), and (110) planes of the hexagonal V_2C , which consists with the XRD results. The HRTEM image of V_2C (bottom-right in Fig. 3b) clearly exhibits the lattice fringe spacing for (100) plane of hexagonal V_2C ($d_{100} = 0.251$ nm), which further proves that V_2C was prepared. The TEM image of the as-prepared nanocrystalline Mo_2C is shown in Fig. 3c. The average size of the primary particles is 30 nm. The SAED rings shown in the inset of Fig. 3c are consistent with the XRD pattern of Mo_2C . Fig. 3d is a HRTEM image of Mo_2C . The regular spacing of the lattice planes is 0.26 nm, which is in agreement with (100) plane of the Mo_2C .

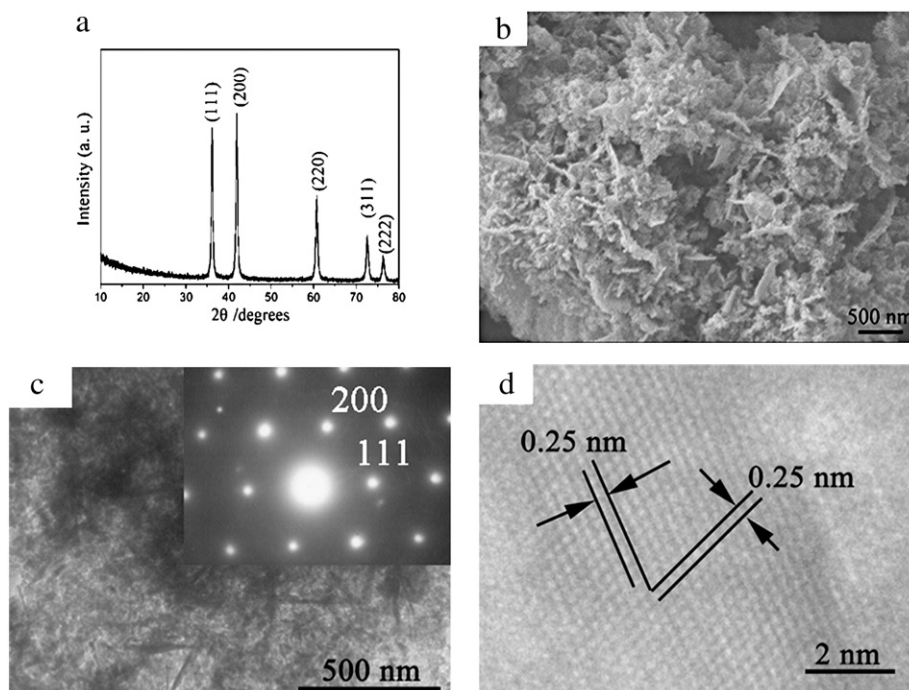


Fig. 1. (a) XRD patterns of as-prepared TiC; (b) FE-SEM pattern of as-prepared TiC; (c) TEM of the as-prepared TiC. The inset shows the corresponding SAED patterns; (d) HRTEM image of the as-prepared TiC.

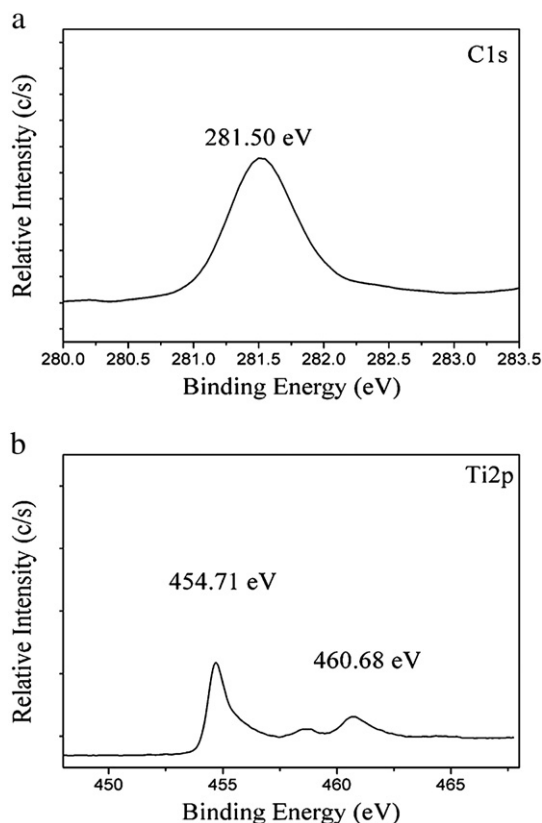


Fig. 2. XPS spectra of the as-prepared TiC (a) C1s region, (b) Ti2p region.

3.3. Characterization of the synthesized TiN, VN and CrN

Transition metal oxides such as TiO_2 , V_2O_5 , and Cr_2O_3 can be converted into related nitrides by this route at 550 °C. X-ray diffraction was used to verify the crystal structures and the phase purity of the materials. The XRD pattern of the synthesized TiN was shown in

Fig. 4a. The XRD pattern indicates that the synthesized TiN has a NaCl-type structure with lattice constant of $a = 4.240 \text{ \AA}$, consistent with the reported value of cubic TiN (JCPDS card, No. 38-1420 $a = 4.241 \text{ \AA}$). No other peaks of impurities of metals or metal oxides can be found from the patterns. The morphologies of the prepared samples were investigated by TEM. The TEM image of the TiN is shown in Fig. 4b, which indicates the synthesized TiN consists of particles in the range of 20–30 nm in diameter. Moreover, the SAED rings/spots (inset of Fig. 4b,) correspond to the (111), (200), and (220) planes of TiN, which confirm the XRD result. The XRD pattern of the synthesized VN nanocrystals was shown in Fig. 5a. All the strong peaks can be indexed to the NaCl-type structure of VN with lattice parameter $a = 4.137 \text{ \AA}$, which is in good agreement with the reported value $a = 4.139 \text{ \AA}$ (JCPDS card, No. 35-0768). The TEM image of the synthesized VN (Fig. 5b) shows that the size of the nanocrystals is in the range of 20–40 nm. Moreover, the SAED rings/spots (inset of Fig. 5b) can be indexed to the (111), (200), and (220) planes of cubic VN. The XRD pattern of the synthesized CrN nanocrystals, shown in Fig. 6a, is basically the same as that of the cubic CrN phase (JCPDS card, No. 76-2494). The lattice constant calculated from the pattern ($a = 4.120$) is very close to the reported value for CrN ($a = 4.140$). TEM image of the synthesized CrN nanocrystals was shown in Fig. 6b, which shows that the particles are agglomerated, and have an average size of about 40 nm. The SAED rings/spots (inset of Fig. 6b) from inner to outer can be indexed to (111), (200) and (220) planes of cubic CrN, which are consistent with the XRD result.

3.4. The possible formation mechanism of the carbides– taking TiC as an example

The solid state metathesis (SSM) route is an effective method in generating metal carbides and nitrides by using metal halides or metal oxides as the metal sources. The process has the advantages of the relatively low reaction temperature and easy removal of the co-produced salt with water. Furthermore, the metal oxides are inexpensive and easy operation metal source in the SSM route, while it requires higher temperature (600–1200 °C) due to the high kinetic barriers. As we know, solid state metathesis reactions are driven

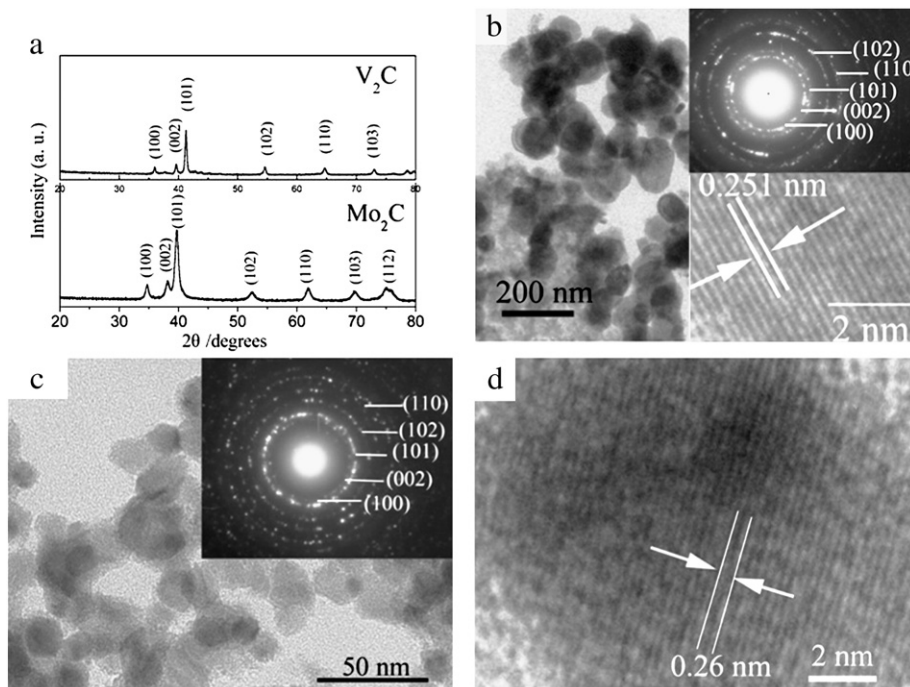


Fig. 3. (a) XRD patterns V_2C sample and Mo_2C sample; (b) TEM image of the as-prepared V_2C sample, the inset shows the corresponding SAED patterns and HRTEM image; (c) TEM image of the as-prepared Mo_2C sample, the inset shows the corresponding SAED patterns; (d) HRTEM image of the as-prepared Mo_2C sample.

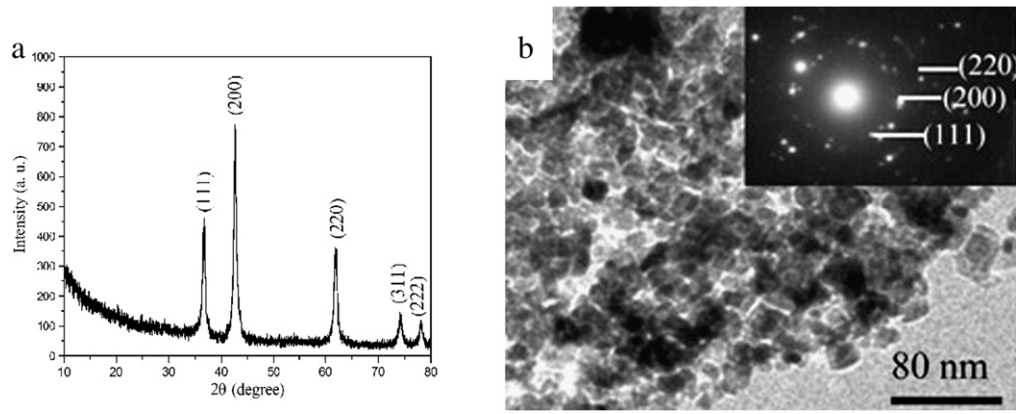


Fig. 4. (a) XRD pattern of the synthesized TiN; (b) TEM images of the synthesized TiN.

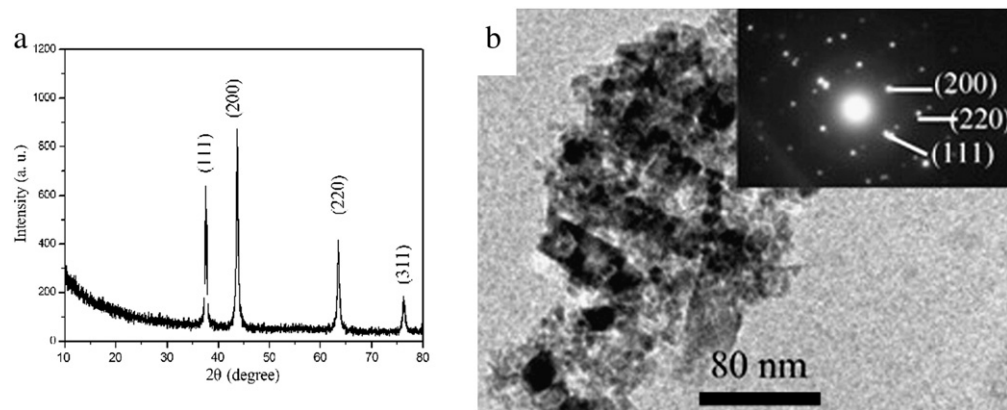


Fig. 5. (a) XRD pattern of the synthesized VN; (b) TEM images of the synthesized VN.

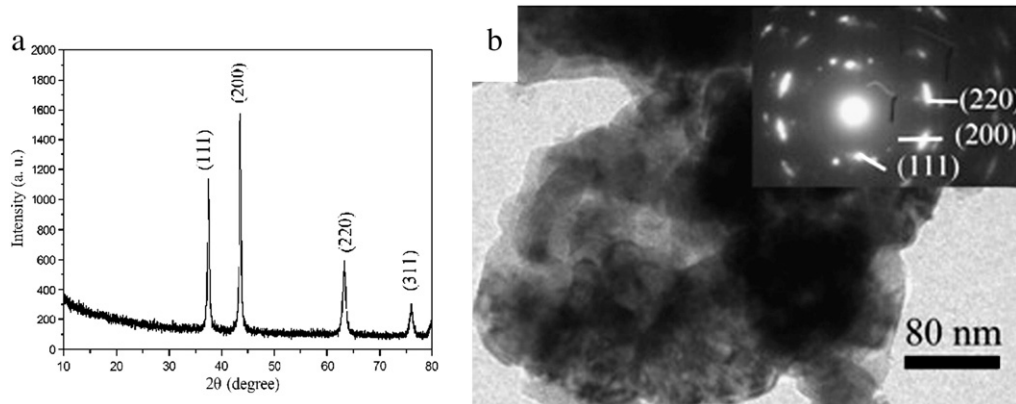


Fig. 6. (a) XRD pattern of the synthesized CrN; (b) TEM images of the synthesized CrN.

primarily by the lattice energy of the coproduced product. The large amount of heat generated in the solid state metathesis reaction urges the reaction to proceed continuously. A range of transition metal carbides were synthesized by the reaction of transition metal oxides with calcium carbide at 1000 °C for 12 h [29]. However, in the present work, we can produce the carbides from metal oxides and calcium carbide with magnesium-assisted at 600 °C. The experimental parameters such as reaction temperature, reactants, and time were investigated to determine the formation mechanism. Taking the formation of TiC as an example, titanium carbide can't be obtained when the temperature was set below 550 °C, while the carbides with good crystallinity were obtained at 600 °C. Under the

present experimental conditions, it was found that only gray powder was obtained by the reaction of TiO₂ and CaC₂. The XRD pattern of the gray powder (Fig. 7a) can be indexed to TiO₂, which indicates that there is no reaction between TiO₂ and CaC₂ at 600 °C. To further understand the formation mechanism of the TiC, a series of time-dependence experiments were carried out. Fig. 7b, c and d show the XRD patterns of the products obtained by the reaction of CaC₂, Mg and TiO₂ for 30 min, 2 h and 4 h, respectively. The XRD patterns (Fig. 7b) can be indexed to the Ti₂O₃ (JCPDS card, No 85-0868), which shows that TiO₂ is reduced to form Ti₂O₃ by magnesium and no carbide (TiC) can be obtained within 30 min. The XRD patterns (Fig. 7c) can be indexed to the mixture of Ti₂O₃ and TiC because

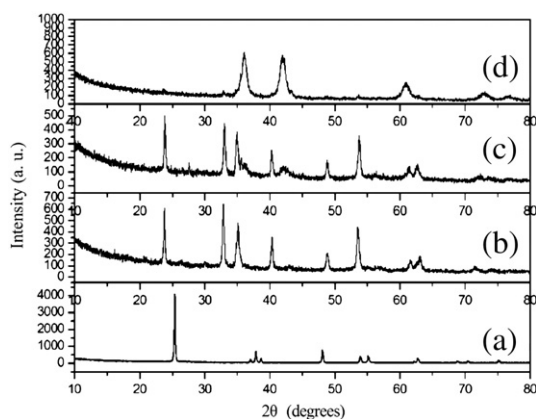
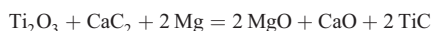


Fig. 7. (a) The XRD pattern of product obtained in the reaction between the CaC_2 and TiO_2 . The XRD pattern of product obtained in the reaction between the CaC_2 , Mg and TiO_2 with different heating time: (b) 30 min, (c) 2 h, and (d) 4 h.

several broad peaks can be indexed to cubic TiC except that of Ti_2O_3 . This result indicates that TiC has been started to form. When the reaction time was prolonged to 4 h, the diffraction peaks of TiC were found in the XRD pattern (shown in Fig. 7d). From the Fig. 7d, the diffraction peaks of Ti_2O_3 can also be found coexisting with those of TiC. The diffraction peaks of the synthesized TiC were very broad in the XRD pattern (shown in Fig. 7d), which indicates that sizes of the obtained TiC are very small. As the reaction time extends from 4 to 10 h, the diffraction peaks in the XRD pattern (Fig. 1a) turn sharper. Based on the above-mentioned experimental facts, the synthesis mechanism has been described by follow processes: Firstly, TiO_2 was reduced to Ti_2O_3 , which is similar to the literature [30]; then, TiC was obtained by the reaction Ti_2O_3 , Mg and CaC_2 . Therefore, the possible reaction could be expressed as follows,



However, the detailed formation mechanism of the metal carbides is also not clear, and the further experiment is underway.

4. Conclusion

Herein, through the magnesium-assisted solid-state reaction, we have successfully prepared metal carbides (TiC, V_2C , Mo_2C) and metal nitrides (TiN, VN, CrN) starting from the corresponding transition metal oxides at 550–600 °C. The FE-SEM image showed that the TiC sample is mainly consisted of flower-like microstructures. TEM image showed the other carbides (V_2C , Mo_2C) and the obtained nitride (TiN, VN, CrN) were consisted of nanoparticles. The possible synthesis mechanism of TiC had been described. This general and simple route may provide new insights into the synthesis of other metal carbides and nitrides.

Acknowledgements

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